

X-Ray Intensity Fluctuation Spectroscopy Observations of Critical Dynamics in Fe₃Al

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We have carried out intensity fluctuation spectroscopy measurements using coherent x rays to study the dynamics of critical fluctuations in a binary alloy at equilibrium. An intense coherent hard x-ray beam, produced from an undulator source, was scattered from a single crystal of Fe₃Al held at temperatures near the B2-DO₃ order-disorder transition. A speckle pattern was observed at the $(\frac{1}{2} \frac{1}{2} \frac{1}{2})$ superlattice reflection. Below T_c it was essentially static, while above T_c it fluctuated in time. The behavior of the normalized time correlation function is consistent with predictions of theory.

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The development of techniques using coherent x rays is of considerable current interest. This is largely because new undulator-based synchrotron x-ray sources are capable of providing coherent hard x-ray beams several orders of magnitude more intense than previously available. In particular, several groups have been working to carry out intensity fluctuation spectroscopy (IFS) measurements using x rays [1]. This “XIFS” technique is especially promising because it should provide one of the few direct measurements of the dynamics of atomic-scale fluctuations. Observations of static speckle patterns in the scattering of coherent x rays from disordered systems have been reported as initial steps toward this goal [1–3]. In this Letter we report our first results using XIFS to measure the dynamics of order fluctuations in a metal alloy near its critical point.

To understand why XIFS is a probe of the dynamics of disorder, it is first necessary to understand the difference between scattering patterns produced by coherent and incoherent beams. When coherent light is scattered from a disordered system, it gives rise to a random diffraction or “speckle” pattern [4]. Such a speckle pattern is uniquely related to the *exact spatial arrangement* of the disorder. In contrast, if the incident beam is not coherent, the speckle is not resolved, and the scattering pattern is determined only by the *probability distribution* of the disorder.

Although the speckle pattern may be difficult to interpret as a spatial structure, a unique advantage in scattering with a coherent beam lies in the ability to observe the *dynamics* of the disorder. When the spatial arrangement of the disorder evolves with time, the speckle pattern also changes. The technique of IFS (also known as dynamic light scattering [5,6]) is simply the observation of the in-

tensity fluctuations at a single point in the speckle pattern to obtain a direct measure of the dynamics. It is commonly used with visible coherent light to study processes such as the diffusion of micron-size particles in liquids. By instead using coherent hard x rays ($\lambda \sim 1 \text{ \AA}$) to conduct XIFS measurements, it should be possible to probe the dynamics of processes involving atomic length scales in a wide variety of materials. In contrast to conventional time-resolved x-ray scattering using an incoherent beam [7], XIFS could be used to study the dynamics of disorder in systems *at equilibrium*, where the exact arrangement of disorder evolves in time but its average distribution does not. In this work we introduce XIFS to study the dynamics of short-range order (SRO) fluctuations associated with a simple phase transition.

Our experiments were conducted at the undulator beam line 9 (Troika) of the European Synchrotron Radiation Facility. Figure 1 is a schematic of the apparatus. The nominal size of the undulator source is 0.21 mm \times 0.97 mm (vert \times horiz) full width at half maximum (FWHM), although slits located 27 m downstream from the source were used to restrict the effective horizontal

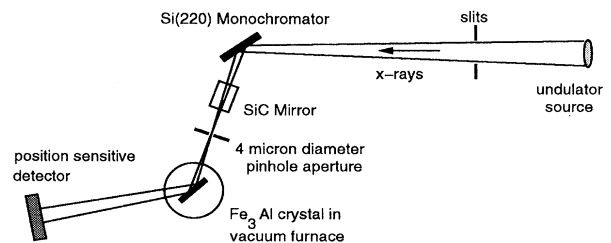


FIG. 1. Schematic of the experimental apparatus.

source size to be equal to that in the vertical. X rays of wavelength $\lambda = 1.05 \text{ \AA}$ and wavelength spread $\Delta\lambda/\lambda = 6 \times 10^{-5}$ were selected by a Si(220) monochromator, diffracting horizontally in the symmetric Bragg geometry. To suppress the harmonic content of the beam, a flat, vertically reflecting SiC mirror was placed downstream of the monochromator. Following the mirror a $4 \mu\text{m}$ diameter pinhole in a $50 \mu\text{m}$ thick platinum foil served as an x-ray aperture. In this geometry, the divergence of rays passing through the pinhole is determined by the source size $A = 0.21 \text{ mm}$ divided by the pinhole-to-source separation $R = 45 \text{ m}$. For radiation passing through the pinhole, the transverse coherence length is $d_t = \lambda R/2A \approx 11 \mu\text{m}$, so that the $4 \mu\text{m}$ pinhole selects a beam which is transversely coherent in both the vertical and horizontal. The flux through the pinhole was 5.7×10^6 photons per second at a typical storage ring current of 100 mA .

The single crystal we studied was slightly off-stoichiometric Fe_3Al (27.1 at. % Al) [8], which has a continuous $B2\text{-DO}_3$ order-disorder transition with critical temperature T_c near 824 K . This transition is analogous to a simple antiferromagnetic Ising model with a nonconserved order parameter (model A dynamics [9]). The specimen was located as close to the pinhole as the sample vacuum furnace would permit (8 cm), in order to be within the near field of diffraction from the pinhole. The sample had a (110) surface normal and was oriented for the $(\frac{1}{2} \frac{1}{2} \frac{1}{2})$ superlattice reflection. The horizontally diffracted beam was observed with a Xe- CO_2 position-sensitive proportional counter. The detector was situated 3.64 m downstream from the specimen and masked by a $95 \mu\text{m}$ ($26 \mu\text{rad}$) high slit. For each detected photon, both the position and the arrival time were recorded. The 4096 position channels each corresponded to $13.5 \mu\text{m}$ ($3.7 \mu\text{rad}$) along the detector. The resolution of the detector was $149 \mu\text{m}$ ($41 \mu\text{rad}$), and the dark count rate was typically 0.003 counts per second per channel. The signal from a beam monitor located between the pinhole and the sample was used to normalize the data to variations in incident intensity.

In order to perfectly resolve a speckle pattern, the longitudinal coherence length $d_l = \lambda^2/2\Delta\lambda \approx 1 \mu\text{m}$ must exceed $2\mu \sin^2\theta$ for the symmetric reflection geometry [2]. Here $\theta = 9.02^\circ$ is the scattering angle and $\mu = 15 \mu\text{m}$ is the x-ray absorption length in the specimen, giving $2\mu \sin^2\theta \approx 0.8 \mu\text{m}$. For the inclined geometry used here, the minimum d_l is about twice as large. The wavelength spread $\Delta\lambda$ used was thus somewhat larger than ideal, and some loss of resolution of the speckle pattern occurred.

We first describe a typical speckle pattern observed with the sample well below T_c , in the long-range order (LRO) phase. At true equilibrium below T_c , the LRO domain size would have grown to equal the sample size; however, the domain coarsening process becomes increasingly slow as it proceeds, so that the sample typically contains a random arrangement of relatively static LRO domains [10].

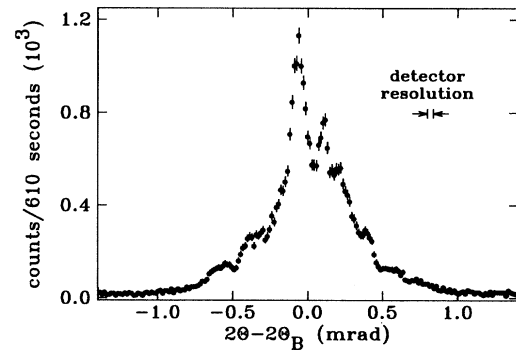


FIG. 2. The $(\frac{1}{2} \frac{1}{2} \frac{1}{2})$ superlattice peak in Fe_3Al at $T = 803 \text{ K}$, showing an imperfectly resolved x-ray speckle pattern. The detector channels have been binned by 4.

By appropriately cooling the sample from above T_c over several hours, we prepared a domain distribution suitable for measuring a static speckle pattern. Figure 2 shows the $(\frac{1}{2} \frac{1}{2} \frac{1}{2})$ superlattice peak at $T = 803 \text{ K}$, 21 K below T_c . The overall width of the peak ($500 \mu\text{rad}$) implies an average domain size of $\sim 200 \text{ nm}$. The structure within the peak is an imperfectly resolved x-ray speckle pattern, formed because the two domain types introduce phase shifts to the scattered radiation which differ by π . The angular size of each “speckle” is expected to be equal to the FWHM of the Fraunhofer diffraction pattern of the pinhole itself, which we measured to be $27 \mu\text{rad}$ [1]. However, because of our $41 \mu\text{rad}$ detector resolution, we expect to measure features with a FWHM of $\sim 50 \mu\text{rad}$, which is in fair agreement with Fig. 2.

For the XIFS measurements, the sample was preannealed at $T = T_c - 2 \text{ K}$ for two days, so that the domain structure was well coarsened and essentially static. The observed superlattice peak FWHM was only $\sim 70 \mu\text{rad}$, or ~ 3 speckles, corresponding to an average domain size of $1.5 \mu\text{m}$. To collect the XIFS data, the sample was heated to successively higher temperatures near T_c . At each point, the sample temperature was controlled to $\pm 0.05 \text{ K}$, and the sample was allowed to equilibrate for about 1 h . The time-averaged peak intensity and resolution-corrected FWHM are shown in Fig. 3, as a function of temperature within $\pm 1 \text{ K}$ of T_c . There is a change in the slope of the intensity data and an abrupt increase in peak width at $T_c = 823.8 \text{ K}$. Note that the count rate near T_c is only about one count per second per speckle. Above T_c , the scattering persists owing to SRO fluctuations. The observed peak widths imply SRO correlation lengths ξ of about 750 nm , although the time and volume averaged are too small to give accurate values of ξ . (We believe that this is the reason that the FWHM at 823.95 K exceeds that at 824.15 K .)

Figure 4 shows the scattered intensity averaged over $220 \mu\text{rad}$ (or about 8 speckles) centered on the superlattice peak, plotted as a function of time for each of five temperatures. The intensity is essentially constant below

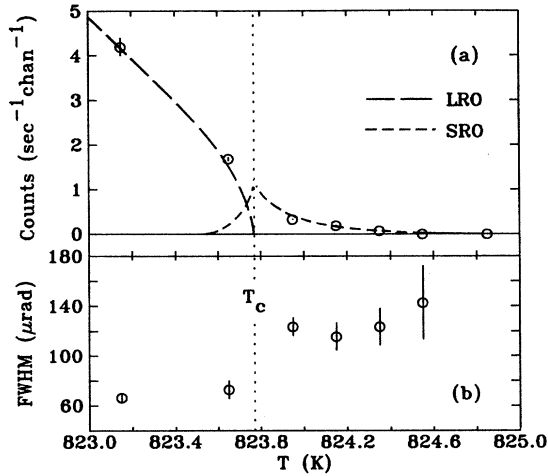


FIG. 3. The peak intensity (a) and overall width (b) of the $(\frac{1}{2}, \frac{1}{2}, \frac{1}{2})$ superlattice peak as a function of temperature. The dashed curves are guides to the eye indicating the LRO and SRO contributions to the scattering. The intensity was normalized to a ring current of 100 mA.

T_c (upper 2 panels), where the signal is dominated by static LRO domains. The observed point-to-point deviations are quantitatively consistent with Poisson counting statistics (indicated by the $\pm 1\sigma$ error bars). This indicates that the source, optics, specimen, and detector meet

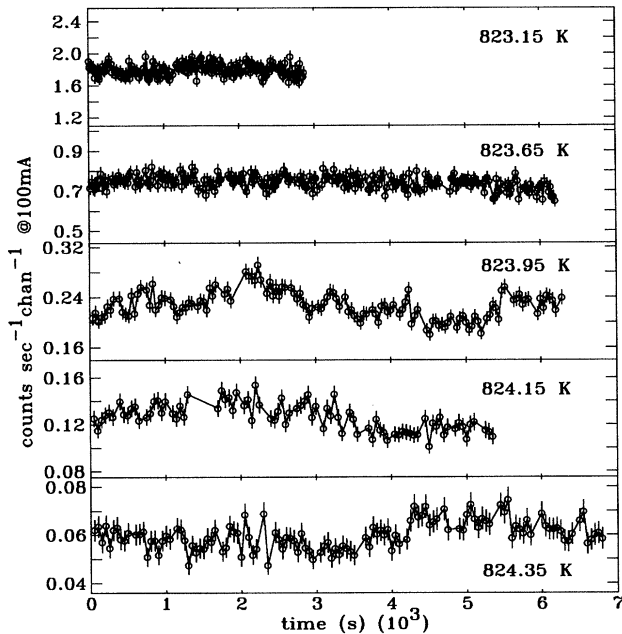


FIG. 4. The scattered intensity averaged over $220 \mu\text{rad}$ at the center of the superlattice peak, plotted as a function of time at various temperatures within $\pm 1 \text{ K}$ of T_c . To facilitate comparison, the data in each panel are plotted on the same scale relative to their mean intensity.

the exacting stability requirements for an XIFS measurement. In contrast, intensity fluctuations of up to 30% relative amplitude are evident above T_c (lower three panels), where the signal is due to SRO critical fluctuations. These are much larger in amplitude than the noise due to counting statistics, and there is a clear correlation between neighboring points. To our knowledge, these data constitute the first observation of the dynamics of equilibrium critical fluctuations at an alloy order-disorder transition.

The fluctuations can be quantified by the normalized time correlation function [11]

$$g^{(2)}(\tau) - 1 = \frac{\frac{1}{N} \sum_{i=0}^N [I(t_i) - \langle I \rangle][I(t_i + \tau) - \langle I \rangle]}{[\frac{1}{M} \sum_{i=0}^M I(t_i)]^2},$$

where $I(t_i)$ is the number of counts in the i th time-bin [12]. This function is nonzero when there are temporal fluctuations and zero when the signal is static. Figure 5 shows $g^{(2)}(\tau) - 1$ calculated from the data of Fig. 4, where the δ -function contribution from counting statistics (based on the mean count rate) has been subtracted from the $\tau = 0$ point in each data set. A clear distinction can be seen in Fig. 5 between the static LRO signal observed below $T_c = 823.8 \text{ K}$ and the SRO fluctuations above T_c . At the two temperatures below T_c , the correlation functions are nearly zero. At the three temperatures above T_c , the correlation functions are nonzero, with amplitudes that are approximately equal and apparent correlation times of about 1000 s. However, because the observations typically span a time range of only 6000 s, these are not reliable estimates for the true correlation times, but instead only lower limits. Such long correlation times are not unreasonable near T_c in this system. A crude estimate using $\tau_{\text{corr}} \approx \tau_0 [(T - T_c)/T_c]^{-\nu z}$ with $\nu \approx \frac{2}{3}$, $z \approx 2$ [9], and a bare correlation time of $\tau_0 \approx 1 \text{ s}$ (based

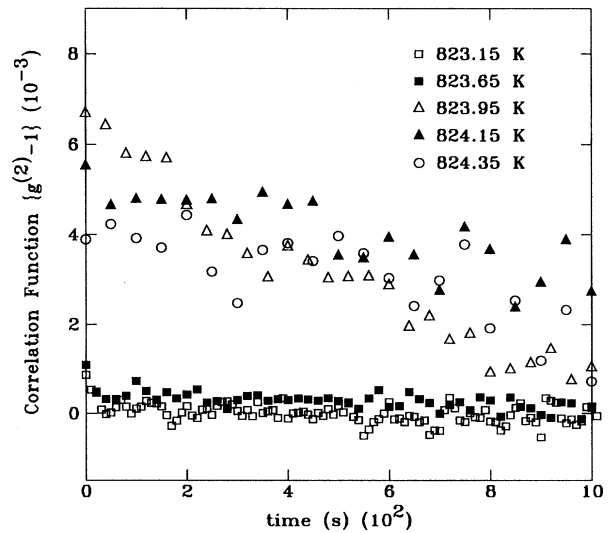


FIG. 5. Normalized time correlation function for the data shown in Fig. 4.

on reequilibration times well above and below T_c [7]), gives $\tau_{\text{corr}} > 10^4$ s at the reduced temperatures used here.

From the theory of dynamic critical phenomena [9] it is straightforward to show [13] that $g^{(2)}(\tau)$ has universal properties near T_c for a system of size $L > \xi$. In particular, the value of $g^{(2)}(\tau = 0) - 1$ at the position of the $(\frac{1}{2}, \frac{1}{2}, \frac{1}{2})$ LRO peak is expected to abruptly change from 2 above T_c to 0 below T_c [14]. In agreement with theory, our data for $g^{(2)}(\tau = 0) - 1$ show an abrupt change from a certain value above T_c to nearly zero below; however, the value observed above T_c is much smaller than 2. The value of 2 is expected when the speckle pattern is perfectly resolved and the time averages are carried out over periods long with respect to the correlation time. In our case, because of the low intensity we have summed the signal over many speckles, reducing fluctuation contrast and thus the observed value of $g^{(2)} - 1$. In addition, the time scale of our measurements may be smaller than the true correlation time, further reducing the observed value because of bias in the determination of $\langle I \rangle$. Experiments with longer counting times and improved intensity and resolution are underway in order to more accurately determine the behavior of $g^{(2)}(\tau)$ near T_c .

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