

Magnetic irreversibility and relaxation in assembly of ferromagnetic nanoparticles

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Measurements of the magnetic irreversibility line and time-logarithmic decay of the magnetization are described for three Fe_2O_3 samples composed of regular amorphous, acicular amorphous, and crystalline nanoparticles. The relaxation rate is the largest and the irreversibility temperature is the lowest for the regular amorphous nanoparticles. The crystalline material exhibits the lowest relaxation rate and the largest irreversibility temperature. We develop a phenomenological model to explain the details of the experimental results. The main new aspect of the model is the dependence of the barrier for magnetic relaxation on the instantaneous magnetization and therefore on time. The time-dependent barrier yields a natural explanation for the time-logarithmic decay of the magnetization. Interactions between particles as well as shape and crystalline magnetic anisotropies define an energy scale that controls the magnetic irreversibility. Introducing this energy scale yields a self-consistent explanation of the experimental data. [S0163-1829(99)03009-X]

I. INTRODUCTION

A ferromagnetic particle becomes monodomain if its size d is reduced below a critical value $d_{\text{cr}} \sim 1-100$ nm, determined by the competition between dipole and exchange energies.^{1,2} Below this critical size, the energy loss due to creation of magnetic domain walls (proportional to d^2) is larger than the gain due to the disappearance of the dipole magnetic-field energy (proportional to d^3). Such monodomain ferromagnetic particles can be viewed as large magnetic units, each having a magnetic moment of thousands of Bohr magnetons. Usually neighboring particles are well separated (10–20 nm), and direct exchange between particles may be neglected. Thus, the magnetic properties of an assembly of nanoparticles are determined by the dipole field energy along with thermal and magnetic anisotropy energies (see, e.g., Refs. 3–6).

Experiments conducted on magnetic nanoparticles show irreversible magnetic behavior below the “irreversibility line” $T_{\text{irr}}(H)$. In particular, the zero-field-cooled (ZFC) and field-cooled (FC) magnetization curves do not coincide, and magnetic hysteresis appears in M vs H curves (see, e.g., Refs. 7–10). Moreover, time-logarithmic magnetic relaxation, towards the thermodynamic equilibrium state, is observed below $T_{\text{irr}}(H)$. Similar observations are reported here for three systems of Fe_2O_3 nanoparticles with different shape and crystalline magnetic anisotropies. These nanoparticle samples were prepared by a sonochemical method, which produces “regular” amorphous nanoparticles.^{11–13} Sonochemical irradiation carried out in the presence of a magnetic field results in synthesis of acicular amorphous nanoparticles.¹³ Annealing of amorphous particles leads to crystallization. In this manner we have prepared regular amorphous, acicular amorphous, and crystalline ferromag-

netic nanoparticles. This enables a study of the effect of shape and crystalline anisotropies on magnetic irreversibility and the relaxation rate. We find that, qualitatively, all three samples exhibit similar irreversible magnetic behavior. However, their irreversibility lines and relaxation rates differ significantly.

Irreversible magnetic behavior similar to that described here is also observed in other systems. A noticeable example is magnetic irreversibility in superconductors.^{14–16} In such systems the origin for irreversibility is the interplay between thermal energy and some energy barrier, which prevents magnetic reorganization in those materials. The microscopic origin of the barrier, however, depends on the system. The magnetic irreversibility in nanoparticles is conventionally associated with the energy required for a particle moment reorientation, overcoming a barrier due to magnetic shape or crystalline anisotropy. It is important to note that the barrier is considered to be independent of the magnetic moment itself.^{8,10,17–32} In superconductors, magnetic irreversibility is due to the inevitable spatial fluctuation of the superconducting order parameter caused by defects, imperfections, etc.; the barrier is the energy required to overcome the pinning due to this disorder.

An important concept in the theory of irreversible magnetic properties of superconductors, based on the work of Anderson,^{14–16} is that the effective barrier for magnetic relaxation increases with time. This is because the superconducting shielding current (proportional to the magnetization) decays with time, causing a decrease in the Lorentz force which drives the fluxons away from their positions. Lottis, White, and Dahlberg³³ have put forward similar arguments to study slow dynamics. They noticed the close analogy between ferromagnetic assemblies and superconductors. Analyzing the results of numerical computations they concluded

TABLE I. Characteristic parameters of the samples.

Sample	d (nm)	Surface area (m^2/g)
Regular amorphous	~ 50	148
Acicular amorphous	$\sim 5 \times 50$	164
Crystalline	~ 100	88

that the decay of the demagnetizing field is the origin of the “quasilogarithmic” relaxation. Although the distribution of particle sizes may explain quasi-time-logarithmic relaxation in a limited time interval, it is not necessary for the explanation of the experimentally observed time-logarithmic relaxation. This approach was later employed in other studies of the magnetic relaxation, for example, in thin magnetic films.³⁴

In this work we adopt the concept of a time varying barrier and derive the phenomenological model to explain magnetic irreversibility and logarithmic magnetic relaxation in nanoparticles. The physics for the time dependence of the barrier in nanoparticles is related to the fact that the effective barrier for reorientation of the magnetic moment of each nanoparticle depends on the internal magnetic field, which includes the average dipole field from surrounding nanoparticles. This average dipole field decreases with time due to the increase of randomness in the orientation of the magnetic moments of the surrounding nanoparticles. This, in turn, causes the increase of the effective barrier with time, yielding a natural explanation to the experimental observation of time-logarithmic relaxation, and a sample-dependent irreversibility line and relaxation rate. Interactions between particles as well as shape and crystalline magnetic anisotropies define an energy scale that controls the magnetic irreversibility. Introducing this energy scale yields a self-consistent explanation of the experimental data.

This article is organized in the following way. In Sec. II we describe the preparation of the three nanoparticle systems. We then describe our experimental results of irreversible magnetic properties at various temperatures, fields, and times. In Sec. III we describe our phenomenological model and derive equations for the irreversibility line and the magnetic relaxation. In Sec. IV we compare the predictions of our model with the experimental results.

II. EXPERIMENTAL

A. Sample preparation and characterization

Three Fe_2O_3 samples composed of regular amorphous, acicular amorphous, and crystalline nanoparticles were prepared by a sonochemical method.^{11–13} For the ultrasound irradiation we used VC-600 Sonics and Materials sonicator with Ti horn at 20 kHz and 100 W cm^{-2} . In Table I we summarize their features. One molar solution of $\text{Fe}(\text{CO})_5$ in decaline was sonochemically irradiated for 3 h in ambient pressure at 0°C . The powder obtained was centrifuged, washed repeatedly with dry pentane (6–7 times, 8500 rpm), and dried in vacuum at room temperature for 3 h. The material obtained has been accumulated from 2–3 sonications and the total amount of Fe_2O_3 was mixed to ensure the reliability of the results. Then, in order to remove organic resi-

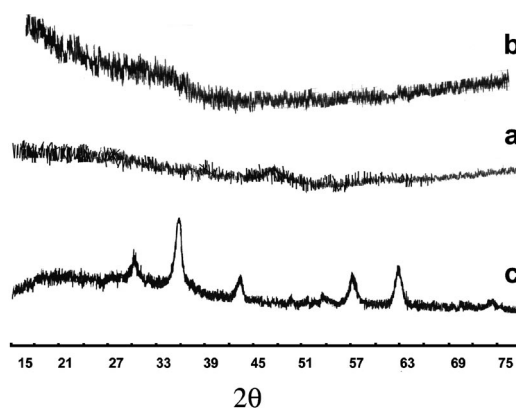


FIG. 1. X-ray diffraction patterns for (a) regular amorphous, (b) acicular amorphous, and (c) crystalline samples.

due, material was annealed in vacuum at $140\text{--}150^\circ\text{C}$ for 3 h. Heating up to this temperature was necessary to evaporate residua of solvents, particularly decaline which has a high boiling point ($189\text{--}191^\circ\text{C}$). The amorphous nature of the particles is confirmed by x-ray diffraction, differential scanning calorimetry (DSC) analysis, and electron-diffraction patterns at selected areas as shown in Figs. 1(a), 2(a), and the inset to 3(a), respectively. The absence of Bragg peaks in Fig. 1(a) demonstrates the absence of the long-range order in the atomic structure; the large endothermic peak in Fig. 2(a) indicates an amorphous to crystalline transition at $\sim 300^\circ\text{C}$. The electron-diffraction pattern of the inset to Fig. 3(a) also confirms the amorphous nature of the particles. A typical particle size of $\sim 50 \text{ nm}$ is inferred from the transmission electron microscopy (TEM) picture of Fig. 3(a).

Acicular amorphous particles have been prepared by performing sonication in external magnetic field of 7 kG for 3 h. The sonication has been carried out in the $0.25M$ solution of $\text{Fe}(\text{CO})_5$ in a flask open to air. We then repeat the wash and dry procedure as described above. The amorphous nature of the particles was confirmed by x-ray diffraction, DSC, and selected area electron-diffraction patterns as shown in Figs. 1(b), 2(b), and the inset to 3(b), respectively. A typical particle length of $\sim 50 \text{ nm}$ and diameter of $\sim 5 \text{ nm}$ are inferred from the TEM picture of Fig. 3(b).

Heating of amorphous Fe_2O_3 up to $370\text{--}380^\circ\text{C}$ in ambient atmosphere for 3–4 h resulted in *crystalline* $\gamma\text{-Fe}_2\text{O}_3$

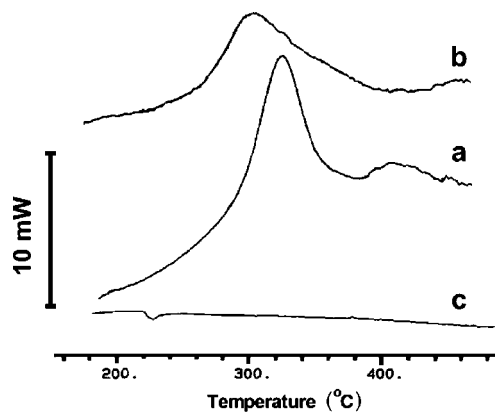


FIG. 2. Differential scanning calorimetry spectra for (a) regular amorphous, (b) acicular amorphous, and (c) crystalline samples.

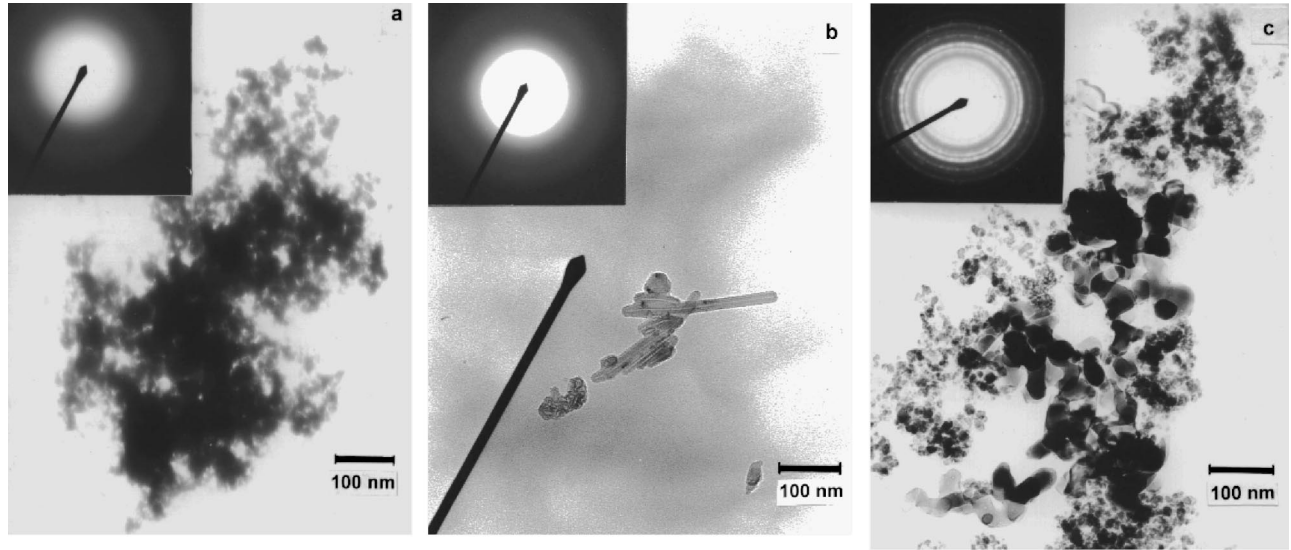


FIG. 3. Transmission electron micrographs for (a) regular amorphous, (b) acicular amorphous, and (c) crystalline samples.

nanoparticles. The nature and the internal structure of the crystalline iron oxide were determined using the x-ray diffraction shown in Fig. 1(c). The DSC data, Fig. 2(c), do not show any endothermic peak. The TEM image of Fig. 3(c) show particles of mean size of ~ 100 nm.

The second column in Table I summarizes typical particle size for the three samples. The third column includes the total surface area of the particles, as measured by Brunauer-Emmett-Teller absorption using N_2 gas as absorbent.

B. Magnetic measurements procedure

A *Quantum Design* MPMS superconducting quantum interference device magnetometer was used for all magnetic measurements reported here. The irreversibility line was determined from ZFC and FC magnetization measurements. Before taking a data point, temperature was stabilized with 0.05 K accuracy and a 30 s pause was sustained. The temperature at which ZFC and FC merge for a constant field H is defined as irreversibility temperature $T_{irr}(H)$. We define the merging point using a criterion $|M_{FC} - M_{ZFC}| \approx 0.1$ emu/g.

The procedure for measurements of magnetic relaxation at different temperatures is as follows: The sample is cooled in $H = 2$ T from room temperature [larger than $T_{irr}(2$ T)] to a target temperature T , the magnetic field is then reduced to 500 G and the magnetic moment is measured for approximately 2 h. The first data point is taken approximately 2 min after the field change.

The field dependence of the magnetic relaxation rate is measured at $T = 20$ K. At this temperature the field is ramped up to $H = 2$ T and reduced back to a target field H , from where the measurements start. The same has been repeated for negative field $H = -2$ T with a consequent increase of the magnetic field to a target value.

C. Results

The experimental results in this section are organized as follows: we first show $M(T, H = \text{const})$ data, and related measurements of magnetic relaxation at different temperatures. From the merging point of the ZFC and FC magneti-

zation curves we extract the irreversibility line for the three samples. From the relaxation measurements we deduce the relaxation rate, as a function of temperature, for the three samples. We then present measurements of magnetization loops $M(T = \text{const}, H)$ and magnetic relaxation at different values of external field. The relaxation rate, as a function of field, is then deduced for the three samples.

Figure 4 exhibits typical results of ZFC-FC magnetization curves and magnetic relaxation at 500 G for the sample composed of amorphous round nanoparticles. The vertical lines of open circles in Fig. 4 depict the relaxation measurements at different temperatures. The vertical arrow indicates the direction of the time increase. The magnetic moment relaxes towards the equilibrium moment M_{rev} , determined by the FC curve. In the inset to Fig. 4 we zoom out at the ZFC-FC curves and indicate by an arrow, the experimental definition of T_{irr} .

The magnetic relaxation data of Fig. 4 are replotted in Fig. 5 as a function of time. The solid lines in Fig. 5 are linear fits for $M \propto \ln(t)$. A qualitatively similar time-logarithmic decay is also observed in the other two samples. Quantitative differences will be discussed below.

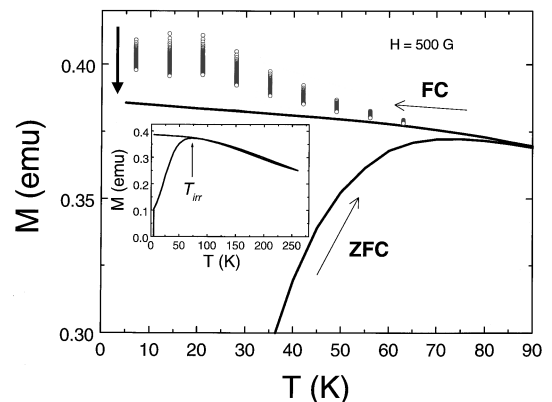


FIG. 4. Typical ZFC-FC curves with superimposed relaxation for the amorphous sample measured at 500 G at different temperatures. *Inset*: full-range ZFC-FC curve.

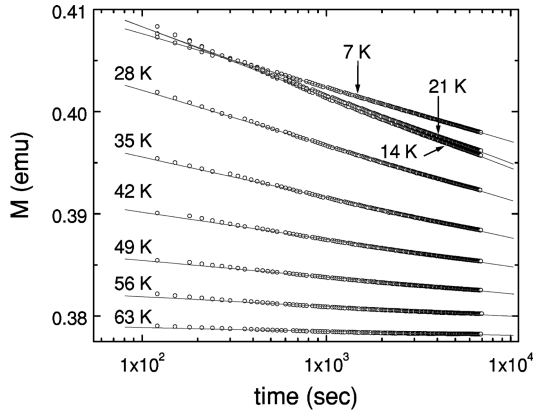


FIG. 5. Typical relaxation curves measured in the amorphous sample in 500 G at different temperatures.

We define the “normalized relaxation rate” $R = |\partial M / \partial \ln(t)| / M_c$, i.e., the logarithmic slope of the relaxation curve normalized by the magnitude of the irreversible magnetization at which the relaxation starts, $M_c = M_0 - M_{rev}$. Here M_0 is the initial value of the total magnetic moment and M_{rev} is the magnetic moment corresponding to a field cooling in 500 G. Figure 6 summarizes the values of R as a function of temperature, for the three samples. At low enough temperatures, R is the lowest for the crystalline sample, intermediate for the acicular amorphous sample and the largest for the regular amorphous sample. Note, that at higher temperatures it looks as if $R(T)$ curves will cross. This is due to a large difference in the absolute values of T_{irr} (90, 162, and 216 K at 500 G for regular amorphous, acicular amorphous, and crystalline, respectively). As shown in Fig. 7, $M(T)$ curves scale with T_{irr} and, therefore, in the inset to Fig. 6 we plot R vs T/T_{irr} . In this presentation, the whole $R(T/T_{irr})$ curve of the crystalline sample is lower than that of the acicular amorphous sample and both are lower than the R curve of the regular amorphous sample.

In Fig. 8 we compare the irreversibility lines for the three samples. The largest irreversibility is found in a crystalline sample, intermediate in the sample with acicular particles and the lowest in the regular amorphous sample. We explain these observations below.

Magnetic irreversibility below T_{irr} is also demonstrated by measuring the magnetization loops $M(H)$. As an example,

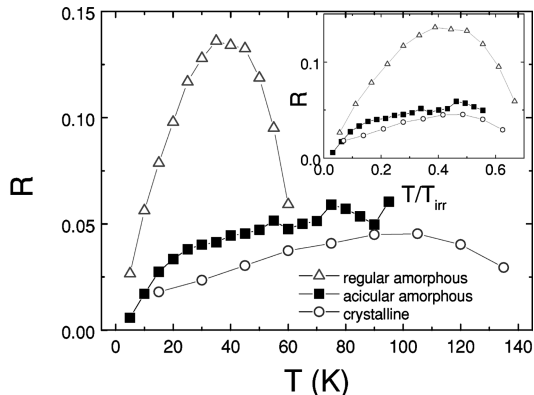


FIG. 6. Normalized logarithmic relaxation rate R for three types of samples as a function of temperature. *Inset*: R as a function of a reduced temperature T/T_{irr} .

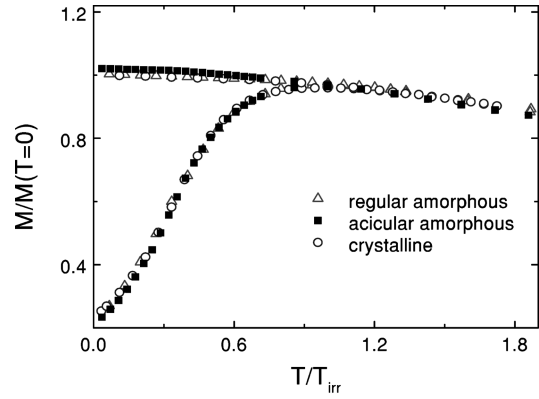


FIG. 7. Scaling of the $M(T)$ FC-ZFC curves with irreversibility temperature.

we show in Fig. 9 $M(H)$ for the amorphous nanoparticles at $T=5$ and 100 K. Magnetic hysteresis is apparent at 5 K, whereas the behavior is purely reversible at 100 K.

The relaxation at different values of the external magnetic field is shown in Fig. 10. The vertical lines represent $M(t)$ curves shown along with the standard magnetization loop. The field dependences of the relaxation rates for our samples are shown in Fig. 11. There is an apparent change in R between low and high fields. At lower fields R is the largest in an amorphous sample, whereas at large fields the relaxation rate in an amorphous sample is the lowest.

III. MAGNETIC RELAXATION IN THE ASSEMBLY OF NANOPARTICLES

A. Time-dependent effective barrier for magnetic reorganization

Magnetic relaxation is a distinct feature of systems with interacting particles, far from thermodynamic equilibrium. In an assembly of ferromagnetic nanoparticles, the elementary process of a change in the magnetization is the rotation of the magnetic moment of a nanoparticle (or cluster of such magnetic moments). In the following we assume that the magnetic anisotropy of each nanoparticle is strong enough to utilize an Ising-like model, i.e., the magnetic moment of each

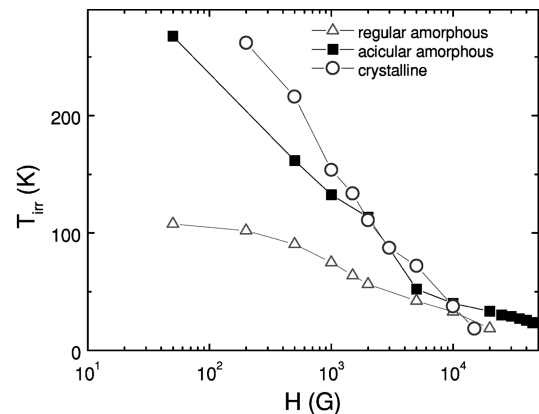


FIG. 8. Irreversibility lines for three types of samples.

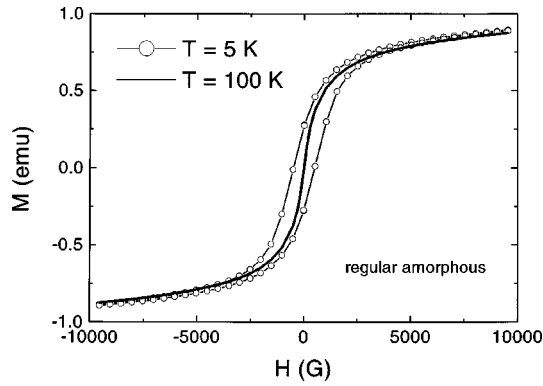


FIG. 9. Typical magnetization loops at $T=5$ K (open circles) and at $T=100$ K (solid line).

particle is aligned only along the anisotropy axis. In Fig. 12(a) we illustrate schematically the orientation of the elementary magnetic moments of several of such nanoparticles. The full arrows represent the size and direction of each magnetic moment. The experimentally measured magnetic moment is determined by the sum of the projections of each individual particle's moment on the direction of the external magnetic field. Note that the directions of the easy axes are randomly distributed. For such a system, the energy W of each magnetic nanoparticle, neglecting for the moment the interparticle interactions, varies with the angle as²¹

$$W = KV \sin^2(\varphi - \theta) - M_p H \cos(\varphi). \quad (1)$$

Here θ is the angle between the easy axis \vec{K} and the external magnetic field \vec{H} , and φ is the angle between the particle magnetic moment \vec{M}_p and the external field. In order to have any magnetic irreversibility and relaxation, the KV term in Eq. (1) must be larger than the $M_p H$ term and we will consider this limiting case. The reduced energy W/KV of Eq. (1) is plotted in Fig. 13 as a function of the angle φ for two different fields $H_1 = 2.5KV/M_p$ (bold) and $H_2 = 0.5KV/M_p$ (light). Since the magnetic anisotropy has no preferable direction, there are two minima in the angular dependence of the energy, as shown in Fig. 13. The external magnetic field fixes the direction of the lowest minima. We denote by U_{12} the barrier for reorientation from the lowest minima (W_1) to

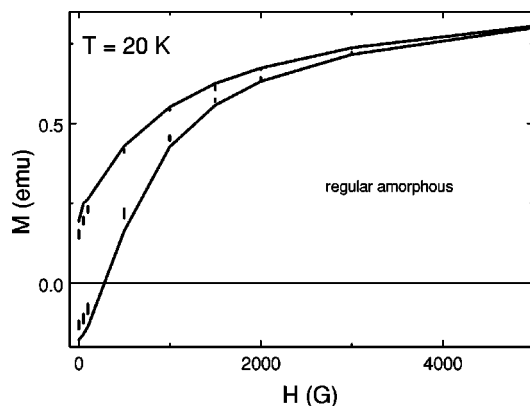


FIG. 10. Magnetic relaxation at different values of magnetic field. Vertical lines are the $M(t)$ curves superimposed on a regular magnetization loop measured at the same temperature.

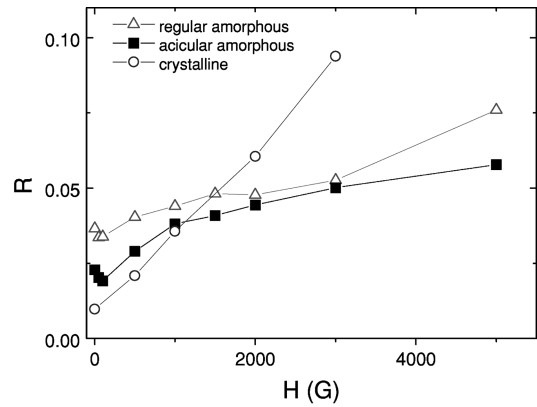


FIG. 11. Normalized logarithmic relaxation rate R for three samples as a function of magnetic field at $T=20$ K.

the other minima (W_2). The backward reorientation requires overcoming the energy barrier U_{21} .

In order to take interparticle interactions into account we view the field H in Eq. (1) as the internal magnetic field, which is the sum of the external field and the dipole field from the surrounding nanoparticles. This local magnetic field depends on the directions of neighboring magnetic moments.³³⁻³⁶ Since the magnetic moment is a statistical average of those moments, the local field depends, on the average, on the total magnetic moment. This induces a feedback mechanism: each reorientation of an individual nanoparticle decreases the total magnetic moment. This is illustrated in Figs. 12(a) and 12(b). Figure 12(a) represents a snapshot of a field-cooled system of nanoparticles in which most of the individual magnetic moments are favorably oriented in a direction such that their projections are along the external field. After a field decrease, as a result of thermal fluctuations, some magnetic moments reorient so that their projection is antiparallel to the external field. The open arrows in Fig. 12(b) represent those reoriented moments.

Since the local dipole field decreases during this process, the average barrier U_{12} increases. As indicated in the Introduction, an increase of the barrier with time is a characteristic of other irreversible systems, such as type-II superconductors in the process of magnetic flux creep.

The dynamics resulting from such a scenario is sketched

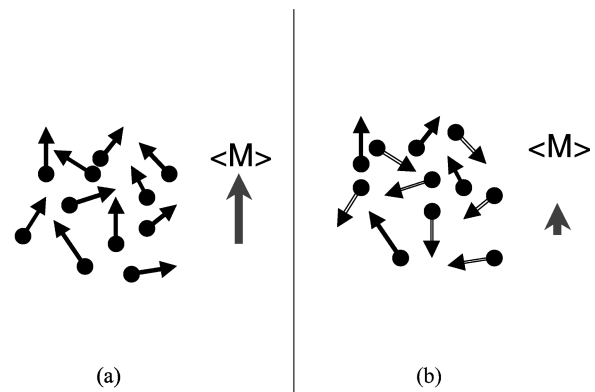


FIG. 12. Schematic snapshots of magnetic moments distribution in powder sample at (a) beginning of the relaxation and (b) at later time.

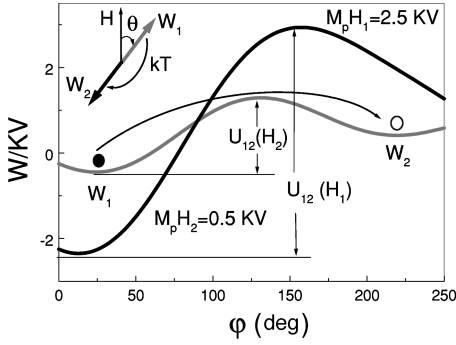


FIG. 13. Energy profiles after FC in high magnetic field (H_1) and after reduction of the magnetic field, when the relaxation starts (H_2).

in Fig. 14. Immediately after reducing the magnetic field, individual magnetic moments are still along the direction of the external field, i.e., in minima W_1 of Fig. 13, as depicted by the dense population of the black dots. During the relaxation process magnetic moments flip to the minimum W_2 in the figure. Since, as discussed above, this barrier depends on the total magnetic moment via dipole fields, it will increase with time as shown in the figure, with dipole fields working on the average against the external field. The total magnetic moment along the magnetic field is thus decreased, as sketched in Fig. 12(b).

B. Equations of magnetic relaxation

In a realistic sample, the directions of easy axes are randomly distributed, the particles cannot physically rotate (e.g., in a dense powder of ferromagnetic nanoparticles), and dipole interactions are strong. We will model this situation as outlined below.

Any given particle i in Fig. 12 has an anisotropy axis at a fixed angle θ_i relative to the external magnetic field. The magnetic moment of this particle is then oriented at an angle φ_i to the field. This angle is defined by the nonlocal energy minimization, due to dipole fields of the surrounding. It is important to note that each particle interacts with a local magnetic field H_i which is the result of a vector sum of the external and dipole fields. At small enough external field and large enough anisotropy φ_i may have two values: $\varphi_i \approx \theta_i$ or $\varphi_i \approx \theta_i + \pi$, which leads to the situation described in Fig. 13, with two energy minima at $W_1^i \approx -M_p H_i \cos(\theta_i)$ and $W_2^i \approx M_p H_i \cos(\theta_i)$. Thermal fluctuations may force a particle moment in the minima W_1^i to change its direction to another minima W_2^i and vice versa. The $W_1^i \rightarrow W_2^i$ rotation requires overcoming a barrier U_{12}^i , and a barrier U_{21}^i for backward rotation, see Eqs. (A1) and (A2) of the Appendix, respectively. We then assume that the field H_i can be represented as a simple sum of the external field H and the collinear to H dipole field H_d (i.e., independent of θ_i). The amplitude of a dipole field H_d at any given site depends upon orientations of the moments of the surrounding particles. If those orientations are totally random (minima W_1 and W_2 are equally occupied) the dipole field is small, whereas if all surrounding particles are in one of the minima the resulting dipole field is maximal. From this simple analysis, we conclude that the magnitude of a dipole field depends upon the total magnetic moment M .

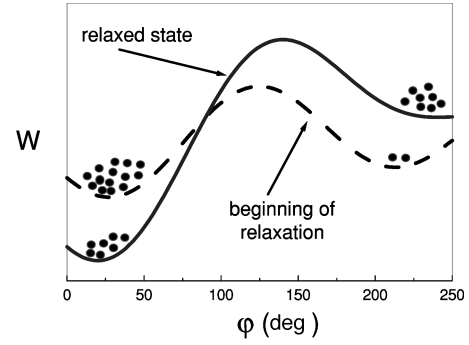


FIG. 14. Energy profiles at the beginning and at the latter stage of the relaxation. Dots indicate population of magnetic moments of the particular energy minima.

Considering the balance of forward and backward rotations, and averaging over the volume of the sample, we show in the Appendix that magnetic relaxation is described by a differential equation similar to that derived for superconductors^{15,16}

$$\frac{\partial M}{\partial t} = -AM_c \exp\left(-\frac{U}{T}\right), \quad (2)$$

where A is an attempt frequency and U is an effective barrier for magnetic relaxation given by

$$U = U_0 \left(1 - \frac{M}{M_0}\right), \quad (3)$$

where

$$U_0 = 2KV + 4M_p(H - \gamma M_{\text{rev}})/\pi \quad (4)$$

and $M_0 = 1/\gamma(\pi KV/2M_p + H - \gamma M_{\text{rev}})$. Here γ is the constant accounting for the strength of the dipole-dipole interactions, M_p is the magnetic moment of an individual particle, K is the anisotropy constant and V is the particle volume. Apparently, as $\gamma \rightarrow 0$ the energy barrier $U \rightarrow U_0$, thus U_0 is the barrier in the assembly of noninteracting particles. It is worth noting that in our model the barrier U depends on the magnetic moment in the same way as that used by Anderson¹⁴⁻¹⁶ for a description of magnetic relaxation in superconductors. In the following we analyze magnetic relaxation described by those equations.

If the barrier for a particle moment reorientation does not depend on the total magnetic moment, i.e., $\gamma=0$ and $U = U_0$, direct integration of Eq. (2) yields

$$M = M_c \exp(-t/\tau), \quad (5)$$

where M_c is the initial irreversible magnetization and $\tau = \exp(U_0/T)/A$ is the *macroscopic* characteristic relaxation time. This result is very similar to that derived in early works for classical Néel's superparamagnetic relaxation, see e.g., Ref. 37. This exponential decay is observed experimentally, for example, in the work of Wegrowe *et al.*¹⁹ on a single nanowire.

If interactions are not negligible, Eq. (2) may be rewritten in dimensionless form:

$$\frac{\partial u}{\partial \tilde{\tau}} = -\exp(-u), \quad (6)$$

where $u = U/T$ and $\tilde{\tau} = t/\tilde{t}$ with

$$\tilde{t} = \frac{M_0}{M_c} \frac{T}{U_0} \frac{1}{A} = \frac{\pi T}{4\gamma A M_p M_c} = \frac{T}{A\Theta}, \quad (7)$$

where we introduced an energy scale Θ , which, as we show below, determines the relaxation process and the irreversibility line:

$$\Theta = M_c U_0 / M_0 = 4\gamma M_c M_p / \pi. \quad (8)$$

This energy is directly related to the strength of the interparticle interactions.

Solving Eq. (6) we obtain

$$u = u_c + \ln\left(1 + \frac{t}{t_0}\right), \quad (9)$$

where $u_c = U_c/T$ is the reduced effective energy barrier at $t=0$, the time when the relaxation starts. The normalization time t_0 is given by

$$t_0 = \tilde{t} \exp\left(\frac{U_c}{T}\right) = \frac{T}{A\Theta} \exp\left(\frac{U_0 - \Theta}{T}\right). \quad (10)$$

Now, using Eqs. (3) and (9) we get the time evolution of the magnetic moment:

$$M(t) = M_c \left[1 - \frac{T}{\Theta} \ln\left(1 + \frac{t}{t_0}\right) \right]. \quad (11)$$

Normalized relaxation rate $R \equiv |\partial M / \partial \ln(t)| / M_c$ is given by

$$R = \frac{T}{\Theta} \frac{t}{t_0 + t}. \quad (12)$$

As we will see below, experiment shows that $t_0 < 1$ s. In our measurements typical time window $\Delta t \approx 100$ s, therefore we can assume $t \gg t_0$ and Eq. (12) predicts that the relaxation rate saturates at $R = T/\Theta$. Thus, measurements of the normalized relaxation rate can provide a direct estimate of the energy scale Θ governing the relaxation process.

C. Irreversibility temperature

The irreversibility temperature T_{irr} of the assembly of magnetic nanoparticles is defined by the condition $M(\Delta t, T_{\text{irr}}) = \Delta M$. Here ΔM is the smallest measured magnetic moment and Δt is the time window of the experiment. Using Eq. (11) we obtain

$$T_{\text{irr}} = \Theta \frac{1 - \Delta M / M_c}{\ln(1 + \Delta t / t_0)} \approx \frac{\Theta}{\ln(1 + \Delta t / t_0)}. \quad (13)$$

thus we can estimate the characteristic time t_0 from measurements of T_{irr} , because the energy Θ can be determined separately from the measurements of the relaxation rate $R \approx T/\Theta$. On the other hand, the irreversibility line $T_{\text{irr}}(H)$ gives the field dependence of Θ . The latter may be obtained also from $R(H)$ measurements. Thus measurements of

$T_{\text{irr}}(H)$ and $R(H)$ in different samples provide a verification of our model on self-consistency.

It is interesting to note that the expression for T_{irr} , Eq. (13), is typical for the blocking temperature of individual noninteracting particles, which is obtained from Eq. (5):

$$T_{\text{irr}}^0 = \frac{U_0}{\ln(\Delta t / t^*)}, \quad (14)$$

where $t^* = 1/(A \ln|M_c/\Delta M|)$ is the characteristic time and U_0 is given by Eq. (4). Energy U_0 is proportional to KV for noninteracting nanoparticles, but it is reduced by a term proportional to γ due to interparticle interactions. This is in agreement with previous works where ‘‘static’’ modifications of the barrier for relaxation were considered.^{17,23,38} Irreversibility temperature, Eq. (14) approaches 0 when $\Delta M \rightarrow 0$, and so does T_{irr}^0 . This reveals an important difference in the physics of the irreversibility line in interacting and noninteracting particles. In the former, there is a true irreversibility in the limit $\Delta M \rightarrow 0$ associated with freezing of magnetic moments due to interparticle interactions. In the case of noninteracting nanoparticles, the apparent irreversibility is due to experimental limitations (finite sensitivity, e.g., ΔM). It is important to stress that this is true only on a macroscopic time scale $\Delta t \gg t^*$, such as relaxation or $M(T)$ measurements. If, however, $\Delta t < t^*$ is realized, for example in Mössbauer measurements, one may detect the irreversibility temperature according to Eq. (14).^{17,24,39}

We also note that in any case T_{irr} is a dynamic crossover from a reversible to an irreversible state and is defined for a particular experimental time window Δt . In the following section we compare our experimental observations with the model developed above.

IV. DISCUSSION

Our phenomenological model provides a description of the irreversible magnetic behavior in the assembly of ferromagnetic nanoparticles. In particular, the model predicts the time-logarithmic decay of the magnetization, see Eq. (11). Also, Eqs. (12) and (13) relate both the irreversibility line and the relaxation rate to a single parameter $\Theta = 4\gamma M_c M_p / \pi$.

The magnetic relaxation data of Fig. 5 reveal, indeed, time-logarithmic relaxation. Fitting these data to Eq. (11) yields the parameters of M_c and Θ . In Fig. 15 we plot the derived energy Θ as a function of temperature for the three samples and find that Θ is the largest for a crystalline sample, intermediate for an acicular amorphous and the lowest for a regular amorphous sample. The straightforward explanation is that in a crystalline sample both γ and M_p are the largest; in an acicular amorphous sample M_p is of the same order as in the regular amorphous, but γ is much larger due to shape anisotropy.

Similarly, we derive the magnetic-field dependence of Θ from the data of Fig. 10 and plot $\Theta(H)$ for three samples in Fig. 16. We note that M_p and γ should not depend on magnetic field. It is therefore expected that the field dependence of Θ is determined by the field dependence of $M_c \approx M_s - M_{\text{rev}}(H)$, which decreases with field, Fig. 10. Figure 16 shows the agreement with this observation. The weak in-

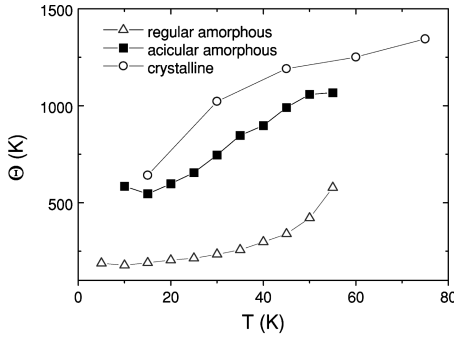


FIG. 15. Temperature dependence of energy Θ extracted from the measurements of normalized relaxation rate for the three samples.

crease of Θ with temperature, Fig. 15, may be related to some nonlinear dependence of barrier U on the magnetic moment.

Independent estimations of Θ are derived from T_{irr} of Fig. 8 using Eq. (13). Comparing Figs. 8 and 16 we get (for three samples) $\Theta/T_{\text{irr}} \approx 4-6$. Thus, $t_0 \approx 0.05-0.5$ s. Note that these values of t_0 are much larger than the “microscopic” values predicted by Néel,³⁷ simply because they reflect collective behavior of the whole assembly controlled by the effective barrier Θ , see Eq. (10), and not a single-particle barrier KV .

Let us now compare the irreversibility lines of different samples, Fig. 8. In most parts of this diagram the region of the irreversible behavior is the largest for a crystalline sample. The amorphous sample containing acicular particles occupies the intermediate space and the amorphous sample embraces the smallest space in this T - H phase diagram. Such behavior is naturally explained in terms of a strength of interparticle interactions, which are the smallest in the case of a regular amorphous sample, intermediate for an acicular amorphous sample (due to shape anisotropy) and the largest for a crystalline sample due to crystalline anisotropy. Also, highest irreversibility temperature of crystalline sample is understood on the basis of its largest particle size.

V. SUMMARY AND CONCLUSIONS

We presented measurements of irreversible magnetization as a function of temperature, time, and magnetic field in three types of ferromagnetic nanoparticles: regular amorphous, acicular amorphous, and crystalline nanoparticles. The results are interpreted using a developed phenomenological approach based on the assumption that the barrier for magnetic moment reorientation depends on the total magnetic moment via dipole fields. This explains the time-logarithmic magnetic relaxation governed by the energy scale Θ related to interparticle interaction. Values of Θ found from measurements of the irreversibility line and the relaxation rate are in perfect agreement, implying the validity of our model.

We demonstrate the influence of a shape anisotropy on magnetic properties of nanoparticles, i.e., irreversible magnetic response of acicular amorphous particles is close to that found in crystalline particles.

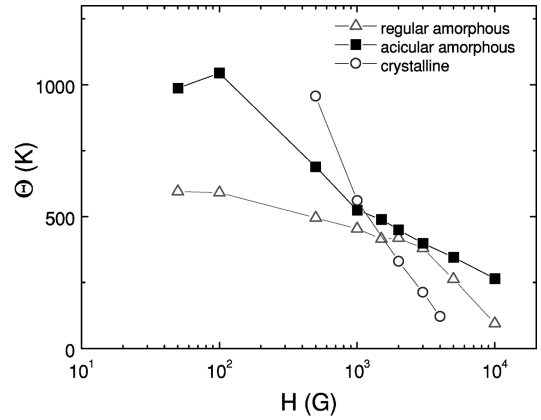


FIG. 16. Magnetic-field dependence of the energy Θ .

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APPENDIX EFFECTIVE BARRIER FOR MAGNETIC RELAXATION AND EQUATION FOR TIME EVOLUTION OF THE MAGNETIC MOMENT

Here we consider in detail the model outlined in the text. We assume that magnetic moment M_p of any given particle i can be in one of the two possible energy minima: $W_1^i \approx -M_p H_i \cos(\theta_i)$ or $W_2^i \approx M_p H_i \cos(\theta_i)$. These minima are separated by the barrier of height $\propto KV + M_p H_i \sin(\theta_i)$. In the presence of thermal fluctuations, a particle moment sitting in the minima W_1^i can spontaneously change its direction to the next minima W_2^i . The energy barrier for such reorientation is

$$U_{12}^i = KV + M_p H_i [\sin(\theta_i) + \cos(\theta_i)]. \quad (\text{A1})$$

The backward rotation is also possible and requires overcoming the barrier:

$$U_{21}^i = KV + M_p H_i [\sin(\theta_i) - \cos(\theta_i)]. \quad (\text{A2})$$

From this point on one can conduct a self-consistent statistical average over angles $\varphi_i(H_i, t, \theta_i)$ in order to evaluate the resulting magnetic moment M of the system. On the other hand, we may try to simplify the problem assuming that the internal field H_i can be represented as a simple sum of the external field H and the collinear to it dipole field H_d (i.e., independent of θ_i). If all easy axes are randomly distributed the average barrier for flux reorientation is then given by

$$U_k \equiv \langle U_i \rangle_k = \frac{1}{N_{ki}} \sum_{i=1}^{N_k} U_i \approx \frac{2}{\pi} \int_0^{\pi/2} U(\theta) d\theta,$$

where $k=12$ or 21 denotes particle's moment flipping from the minima W_1 to the minima W_2 , or backward, respectively. Using Eqs. (A1) and (A2) we find

$$U_{12} = KV + 4M_p H_i / \pi, \quad (\text{A3})$$

$$U_{21} = KV. \quad (\text{A4})$$

Let us now consider a situation where temperature is higher than irreversibility temperature and system is at thermal equilibrium. The number of particles jumping per unit time from one minima to another is proportional to $N_k \exp(-U_k/T)$. The condition for equilibrium is

$$N_1 e^{-U_{12}/T} = N_2 e^{-U_{21}/T}.$$

Thus

$$N_2 = N_1 \exp\left(-\frac{U_{12} - U_{21}}{T}\right) = N_1 \exp\left(-\frac{4M_p H_i}{\pi T}\right).$$

It is clear that the difference $n = N_1 - N_2$ determines the resulting magnetic moment of a system. If the total number of particles in the system is N , the difference n is

$$n = N_1 - N_2 = N \frac{1 - \exp(-4M_p H_i / \pi T)}{1 + \exp(-4M_p H_i / \pi T)} = N \tanh\left(\frac{2M_p H_i}{\pi T}\right). \quad (\text{A5})$$

The total reversible magnetic moment then is

$$M_{\text{rev}} \approx M_p n = M_p N \tanh\left(\frac{2M_p H_i}{\pi T}\right). \quad (\text{A6})$$

This formula is similar to the expression for the Ising superparamagnet and simply reflects the two-state nature of our model.^{37,40} The difference is, however, that the physical magnetic field is the total (external + dipole) field H_i .

Dipole field H_d at any given site depends upon orientations of the moments of the surrounding particles. If those orientations are totally random (minima W_1 and W_2 are equally occupied) the dipole field is small, whereas if all surrounding particles are situated in one of the minima the resulting dipole field is maximal. From this simple picture, we conclude that the magnitude of a dipole field depends upon the total magnetic moment of a sample $M_{\text{rev}} + M$, where M_{rev} is given by Eq. A6 and M is the irreversible, time-dependent contribution to the total magnetic moment resulting from the finite relaxation time needed for a system to equilibrate. Therefore, we may write $H_i = H - \gamma(M_{\text{rev}} + M)$. Here γ is the coefficient accounting for the contribution of dipole fields. Now we can obtain the equation for reversible magnetization from Eq. (A6):

$$M_{\text{rev}} \approx M_S \frac{\tanh(2M_p H / \pi T)}{1 + \gamma(2M_p M_S / \pi T)}, \quad (\text{A7})$$

where $M_S = M_p N$. We note that this formula is valid at small enough fields $2M_p H / \pi < T$ when particle moments are almost locked along the easy axes and small enough interactions (i.e., $H > \gamma M_{\text{rev}}$). The important result is that reversible magnetization decreases as the interparticles interaction increases. Interestingly, Eq. (A7) provides a good description of the experimental data.

Thus, the barriers for moment reorientation in Eqs. (A3) and (A4) can be rewritten as

$$U_1 = KV + 4M_p [H - \gamma(M_{\text{rev}} + M)] / \pi, \quad (\text{A8})$$

$$U_2 = KV. \quad (\text{A9})$$

We shall now consider direct and backward moment rotation processes in a nonequilibrium state. As above, we denote by N_1 and N_2 number of moments in energy minima 1 and 2, respectively. The total number of particles in the system is $N = N_1 + N_2$. The magnetic moment is proportional to the difference $n = N_1 - N_2$. During small time δt this difference changes as

$$\delta n = \left[N_1 \exp\left(-\frac{U_1}{T}\right) - N_2 \exp\left(-\frac{U_2}{T}\right) \right] \delta t. \quad (\text{A10})$$

Using simple algebra and the above relationships between N_1 , N_2 , n , and N , we get

$$\frac{\delta n}{\delta t} = -\exp\left(\frac{U_1 + U_2}{T}\right) \left[n \cosh\left(\frac{U_1 - U_2}{T}\right) + N \sinh\left(\frac{U_1 - U_2}{T}\right) \right]. \quad (\text{A11})$$

From this we arrive at a nonlinear differential equation governing process of magnetic relaxation *not too close to equilibrium*:

$$\frac{\partial M}{\partial t} \approx -A \exp\left(\frac{U_1 + U_2}{T}\right) \left[(M_{\text{rev}} + M) \cosh\left(\frac{U_1 - U_2}{T}\right) + M_s \sinh\left(\frac{U_1 - U_2}{T}\right) \right], \quad (\text{A12})$$

where A is a constant measured in s^{-1} and having the meaning of attempt frequency.

Equation (A12) can be simplified considering magnetic relaxation not too close to equilibrium and retaining our assumption that anisotropy contribution to the magnetic energy is much larger than that of magnetic field (both conditions are better satisfied at low fields). In this case, Eq. (A12) may be approximated in a reduced form:

$$\frac{\partial M}{\partial t} = -AM_c \exp(-U/T), \quad (\text{A13})$$

where M_c is the total magnetic moment at the beginning of relaxation and U is the effective barrier:

$$U = 2KV + 4M_p (H - \gamma M_{\text{rev}} - \gamma M) / \pi = U_0 \left(1 - \frac{M}{M_0} \right), \quad (\text{A14})$$

where $U_0 = KV + 4M_p (H - \gamma M_{\text{rev}}) / \pi$ and $M_0 = 1 / \gamma (\pi KV / 2M_p + H - \gamma M_{\text{rev}})$.

We reiterate that Eq. (A13) is valid only in the case when the magnetic anisotropy is large and magnetic moment is far from equilibrium. Close to equilibrium, one ought to consider Eq. (A12).

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